

Modeling Complex Diffusion Mechanisms in $L1_2$ -Structured Compounds

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Supported by NSF grant DMR 06-06006 (Metals Program)
and KY EPSCoR grant RSF-012-03

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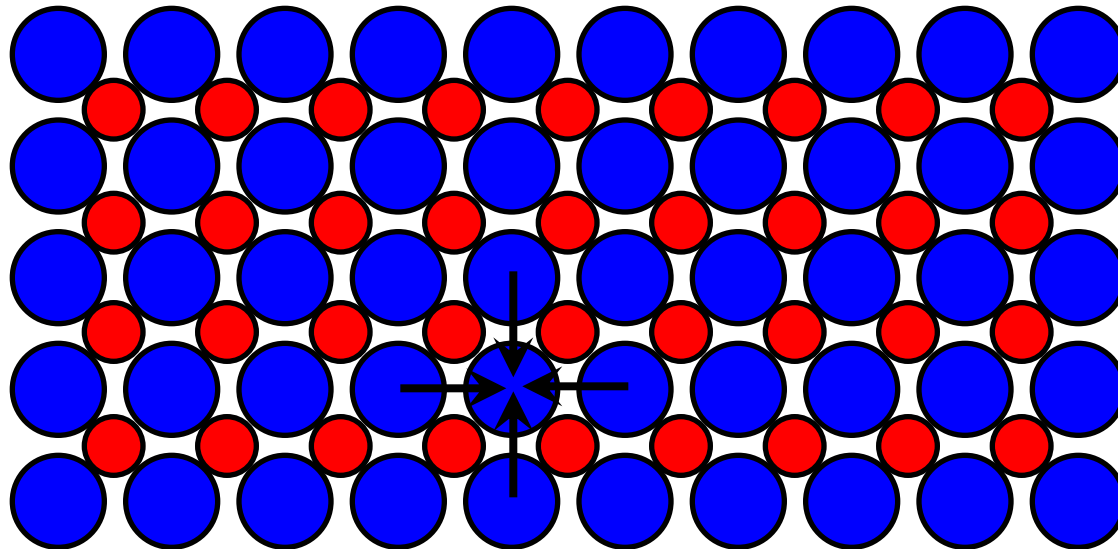
Modeling Complex Diffusion Mechanisms in $L1_2$ -Structured Compounds

Overall goal: To determine when PAC and other hyperfine methods can be used to identify diffusion mechanisms

1. Diffusion mechanisms in intermetallic compounds
2. Recent PAC findings in $L1_2$ -structured compounds
3. Stochastic models of complex mechanisms
4. Simulations for $L1_2$ compounds

Diffusion in intermetallic compounds

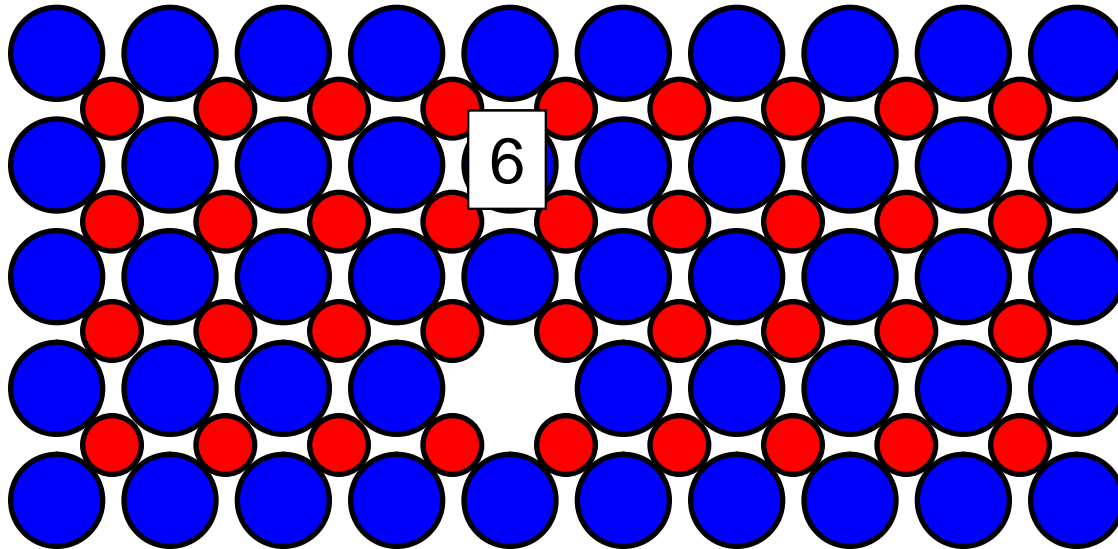
example: simple vacancy diffusion



Diffusion in intermetallic compounds

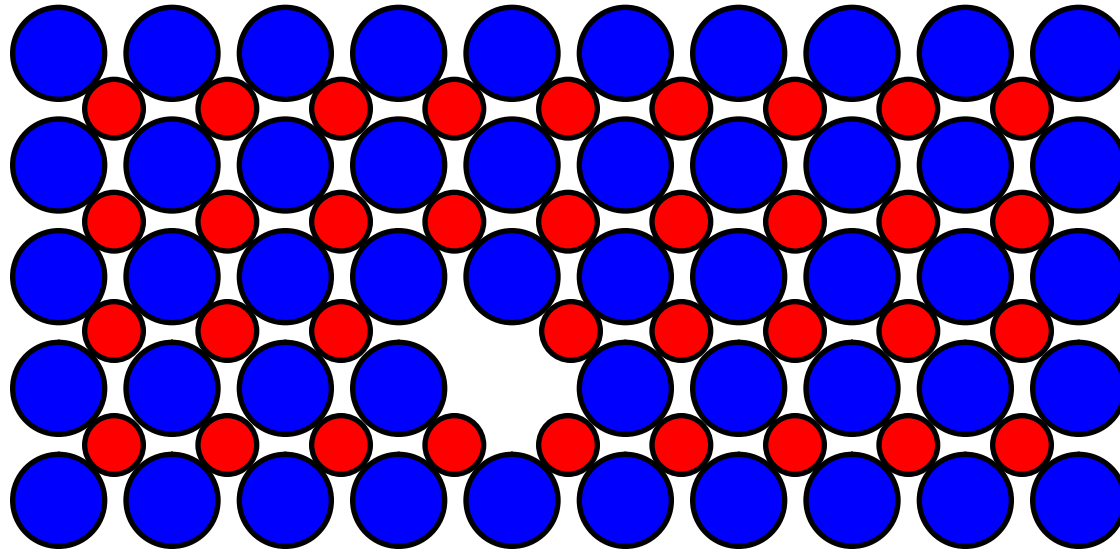
Only first-neighbor jumps allowed

example: 6-jump cycle



Diffusion in intermetallic compounds

Possible alternative: divacancy



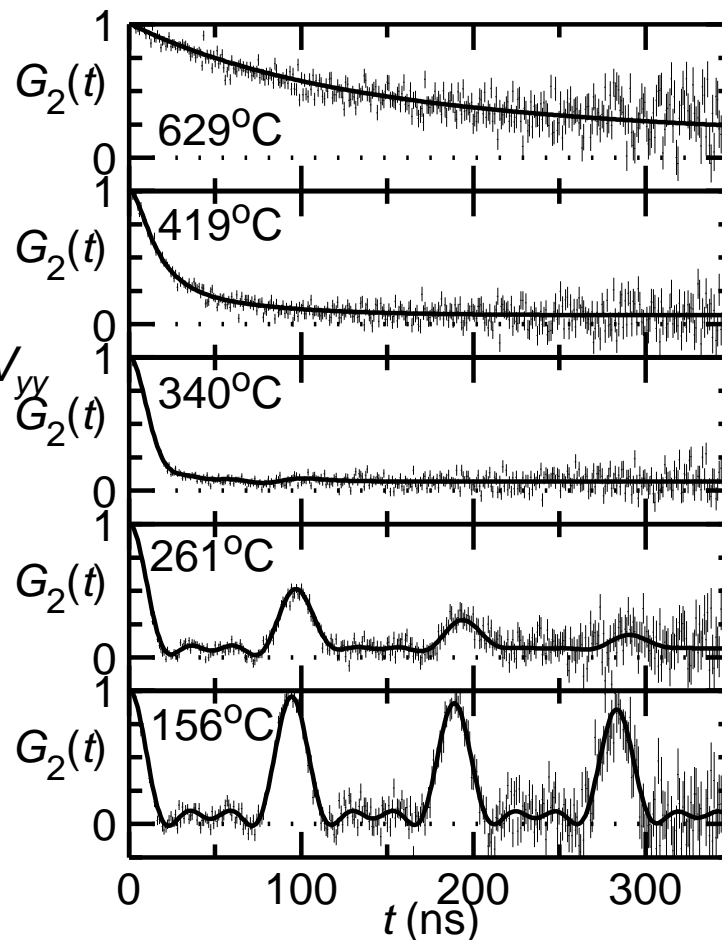
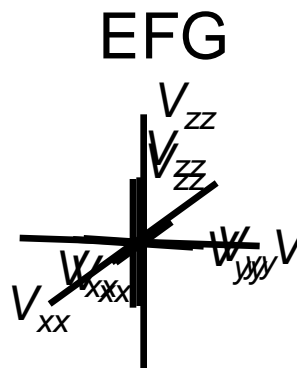
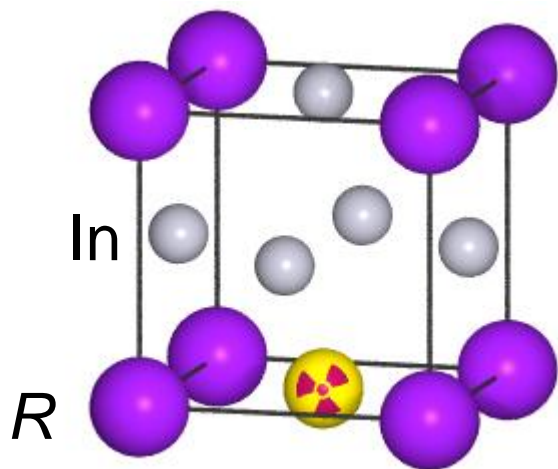
Diffusion in intermetallic compounds

- How to distinguish diffusion mechanisms in alloys (elements A and B) experimentally?
- Conventional radiotracer sectioning methods
 - Measure diffusion coefficients of each element, D_A and D_B and compare ratios.
 - *E.g.* in $A_3B (L1_2)^*$
 - $0.11 < D_A/D_B < 0.85$ for 6-jump cycle
 - $0.07 < D_A/D_B < 3$ for divacancy
- Perturbed angular correlation spectroscopy?

*Koiwa, Numakura, Ishioka, *Def. & Diff. Forum* **143-147**, 209 (1997)

PAC measurements of Cd jump rates in $L1_2$ structured compounds*

- RIn_3 , RGa_3 , RSn_3 with R =rare earth
- $L1_2$ structure:



*Collins, Jiang, Bevington, Selim, Zacate, *PRL* **102**, 155901 (2009) and refs. therein;
 Jiang, Zacate, Collins, *Def. & Diff. Forum* **289-292**, 725 (2009);

PAC measurements of Cd jump rates in $L1_2$ structured compounds

- RIn_3 , RGa_3 , RSn_3 with R =rare earth
- Jump rates at phase boundaries:

jump rate larger for R -poorer boundary	jump rate larger for R -rich boundary
$LaIn_3$, $CeIn_3$, $PrIn_3$ divacancy or 6-jump mechanism	$LuIn_3$, $TmIn_3$, $ErIn_3$, $HoIn_3$, $DyIn_3$, $TbIn_3$, $GdIn_3$ $LaSn_3$, $CeSn_3$, $SmSn_3$, $GdSn_3$

Collins, Jiang, Bevington, Selim,
Zacate, *PRL* **102**, 155901 (2009)

Simple vacancy mechanism

PAC measurements of Cd jump rates in $L1_2$ structured compounds

- RIn_3 , RGa_3 , RSn_3 with R =rare earth
- Jump rates at phase boundaries indicate complex diffusion mechanism in light rare earth tri-indides
- No defects observed in spectra
- Is it possible to observe direct evidence?

436 different binary compounds with $L1_2$ structure

Method of simulating PAC spectra

Stochastic model of Winkler and Gerdau:

$$G_2(\omega) = \sum_q G_q \exp \left[-(\lambda_q + i\omega_q) t \right]$$

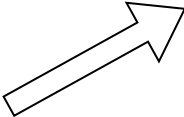
where

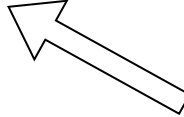
$$G_q = \sum_{m_1 m_2 P} \left(\frac{1}{2I + m_1 + m_2} \begin{pmatrix} I & I & k \\ m_1' & -m_1 & P \end{pmatrix} \begin{pmatrix} I & I & k \\ m_2' & -m_2 & P \end{pmatrix} \right) \sum_{abr} p_a \begin{pmatrix} m_2 m_2' | N_q^r \\ m_1 m_1' \end{pmatrix}$$

The $\begin{pmatrix} \lambda_q + i\omega_q \\ \end{pmatrix}$ and $\begin{pmatrix} | N_q^r \\ \end{pmatrix}$ are eigenvalues and eigenvectors of the Blume matrix

Method of simulating PAC spectra

Blume matrix $\hat{B} = -iH_{st}^{\times} / \hbar + \hat{R}$

Hamiltonians of all accessible EFGs 

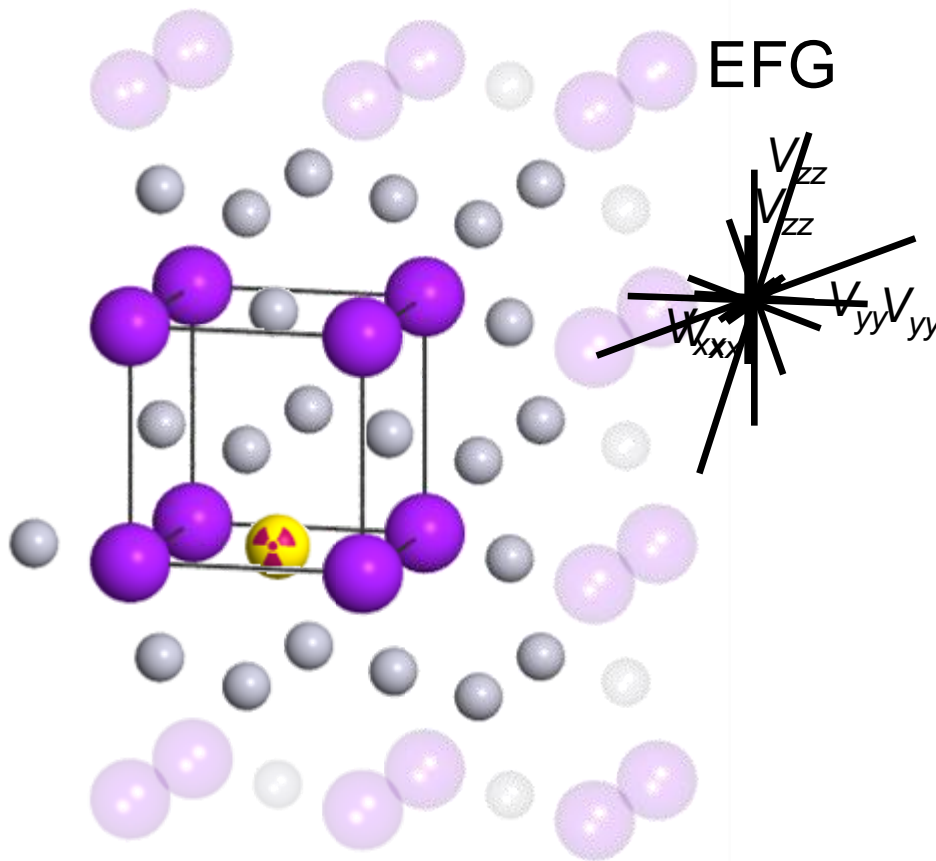
transition rates among EFGs 

- Model completely specified by
 1. List of Hamiltonians for all accessible states
 2. Transition rates among the states
 3. Initial occupations of accessible states
- Use *Stochastic Hyperfine Interactions Modeling Library* (SHIML) to calculate spectra



Simple vacancy diffusion in $L1_2$ compounds

Review of the simple vacancy model*



15 different EFGs
4 transition rates
in the near-neighbor
approximation

*Muhammed, Zacate, Evenson, *Hyperfine Interact.* **177**, 45 (2007)

Simple vacancy diffusion in $L1_2$ compounds

Review of the simple vacancy model*

transition rates

adjacent: $r_a = w_1$

exchange: $r_x = w_2$

detrapping: $r_d = 5[V]w_3$

trap: $r_t = 10[V]w_4$

15 different EFGs

4 transition rates

r_x – exchange

r_a – adjacent

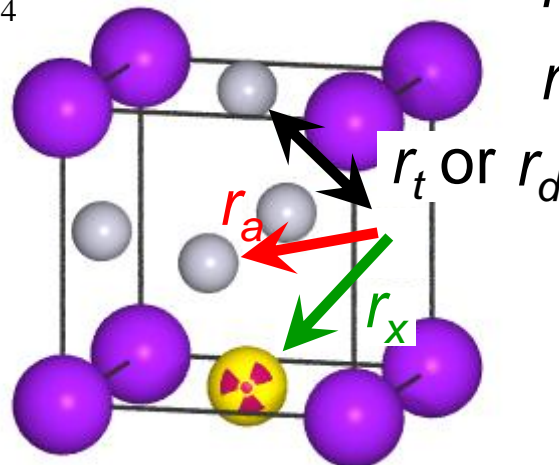
r_d – detrapping

r_t – trap

[V] = vacancy concentration

w_1 - w_4 are from the

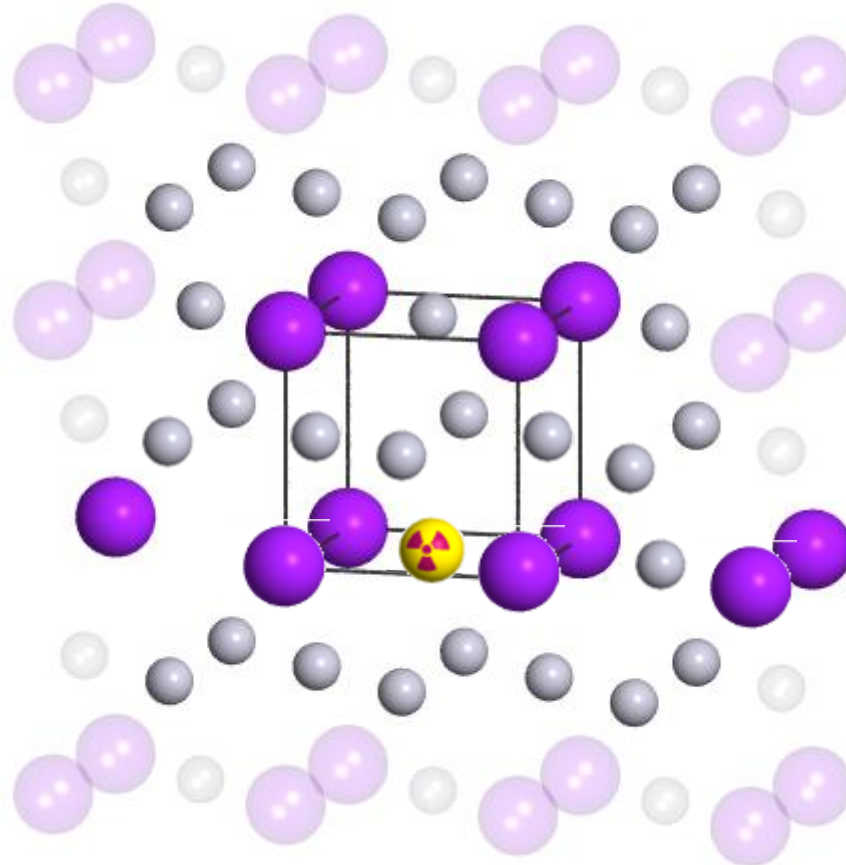
5-frequency model



*Muhammed, Zacate, Evenson, *Hyperfine Interact.* **177**, 45 (2007)

Complex Diffusion in $L1_2$ compounds

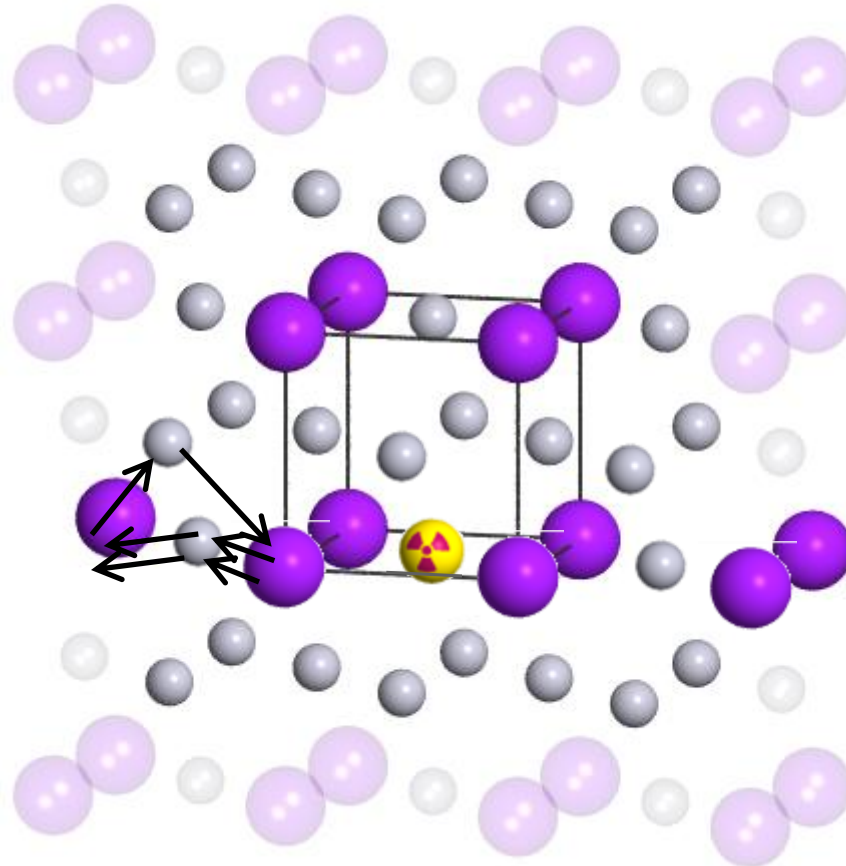
Divacancy in A_3B



45 EFGs
in the near
neighbor
approximation

Simulating Simple Vacancy Diffusion in Cu_3Au -Structured Compounds...

6-jump cycle A_3B



588 EFGs (!)
in the near
neighbor
approximation

Deriving stochastic models linked to defect jump rates

Developed a nearly automated 3 step process

1. Determine all possible configurations of defects near the probe and atomic jumps among them
2. Determine unique EFGs to a cutoff distance from the probe and rates of transition
3. Correct rates for transitions out of defect-free EFGs

Deriving stochastic models linked to defect jump rates

- Step 1
 - Types of defect jumps identified and assigned rates w_k

- Step 2
 - Determine unique EFGs: $k \llbracket q, j \rrbracket$
 - Determine reorientation rate from EFG q to EFG r .

index of j^{th} degenerate configuration with EFG q

$$R_{rq} = \sum_{ij} p_{k \llbracket q, j \rrbracket} W_{k \llbracket r, i \rrbracket \llbracket q, j \rrbracket}$$

0 or a rate w_k

$$p_{k \llbracket q, i \rrbracket} = \frac{P_{k \llbracket q, i \rrbracket}}{\sum_j P_{k \llbracket q, j \rrbracket}}$$

probabilities of defect configurations, i.e. defect concentrations

Deriving stochastic models linked to defect jump rates

• Step 2

- Determine unique EFGs: $k \llbracket q, j \rrbracket$
- Determine reorientation rate from EFG q to EFG r .

$$R_{rq} = \sum_{ij} p_{k \llbracket q, j \rrbracket} W_{k \llbracket r, i \rrbracket \llbracket q, j \rrbracket}$$

index of j^{th} degenerate configuration with EFG q

0 or a rate w_k

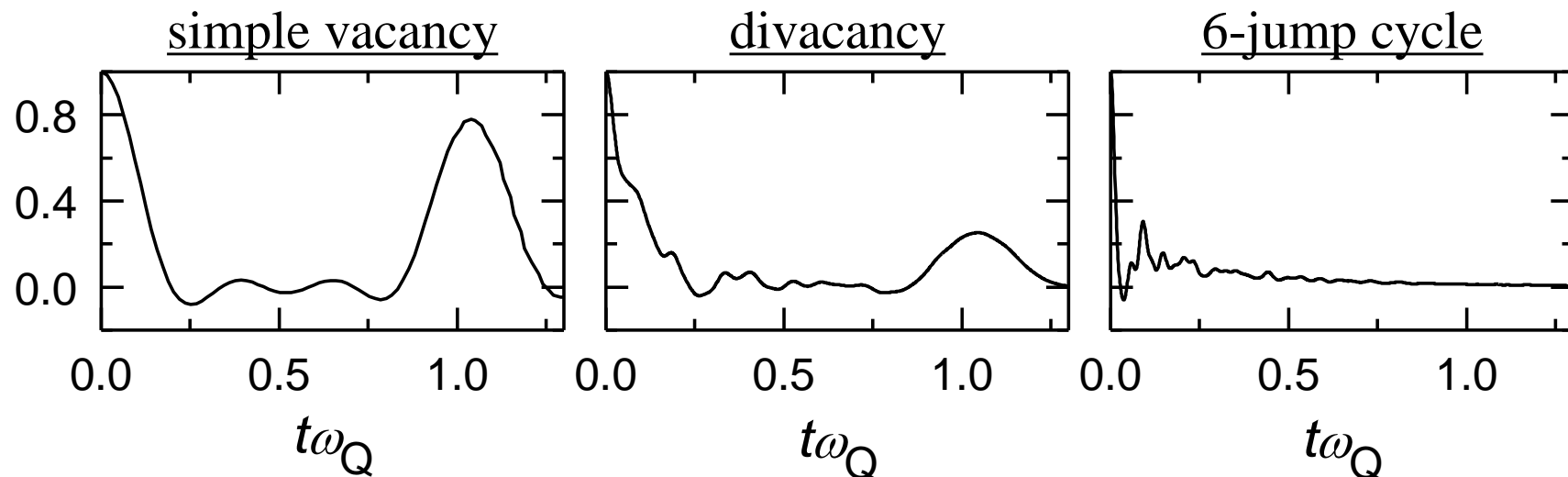
$$p_{k \llbracket q, i \rrbracket} = P_{k \llbracket q, i \rrbracket} / \sum_j P_{k \llbracket q, j \rrbracket}$$

• Step 3

- For q indexing a defect free EFG, $\sum_j P_{k \llbracket q, j \rrbracket} \rightarrow \infty$
so that $R_{rq} = N_{rq} \llbracket \text{defect} \rrbracket w$

number of jump paths \uparrow \uparrow a rate w_k

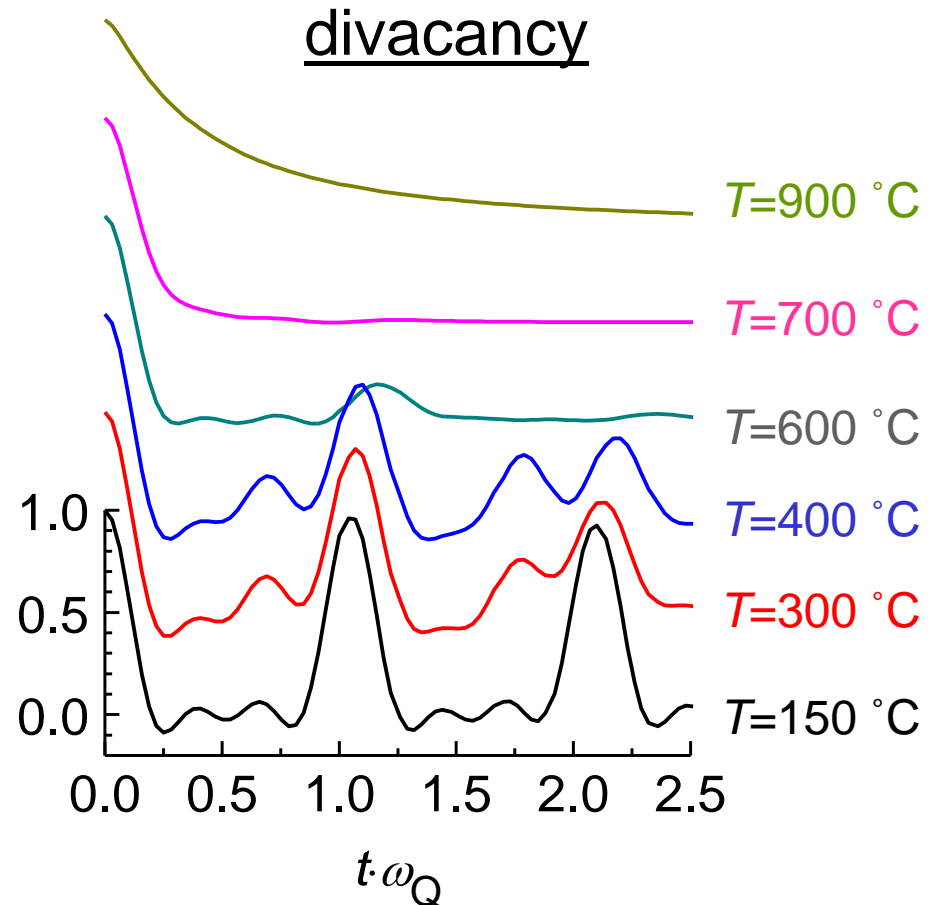
Comparison of diffusion models in $L1_2$



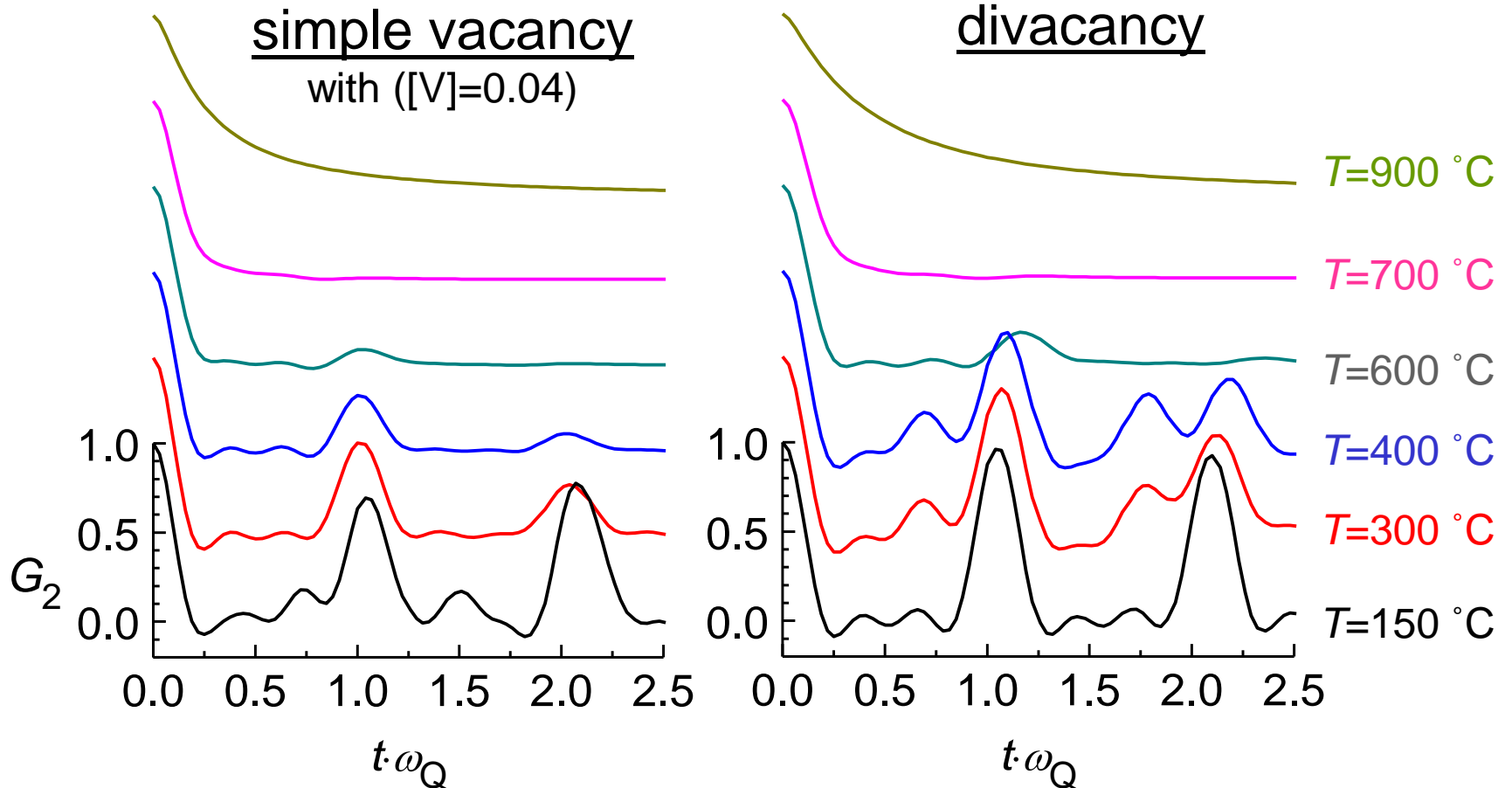
- Defect charges chosen so that contribution of each defect to the EFG is $5.66|V_0|$ where $|V_0|$ is the strength of the lattice EFG.
- All jump rates = ω_Q
- All [defect] = 0.01

Comparison of diffusion models – part 2

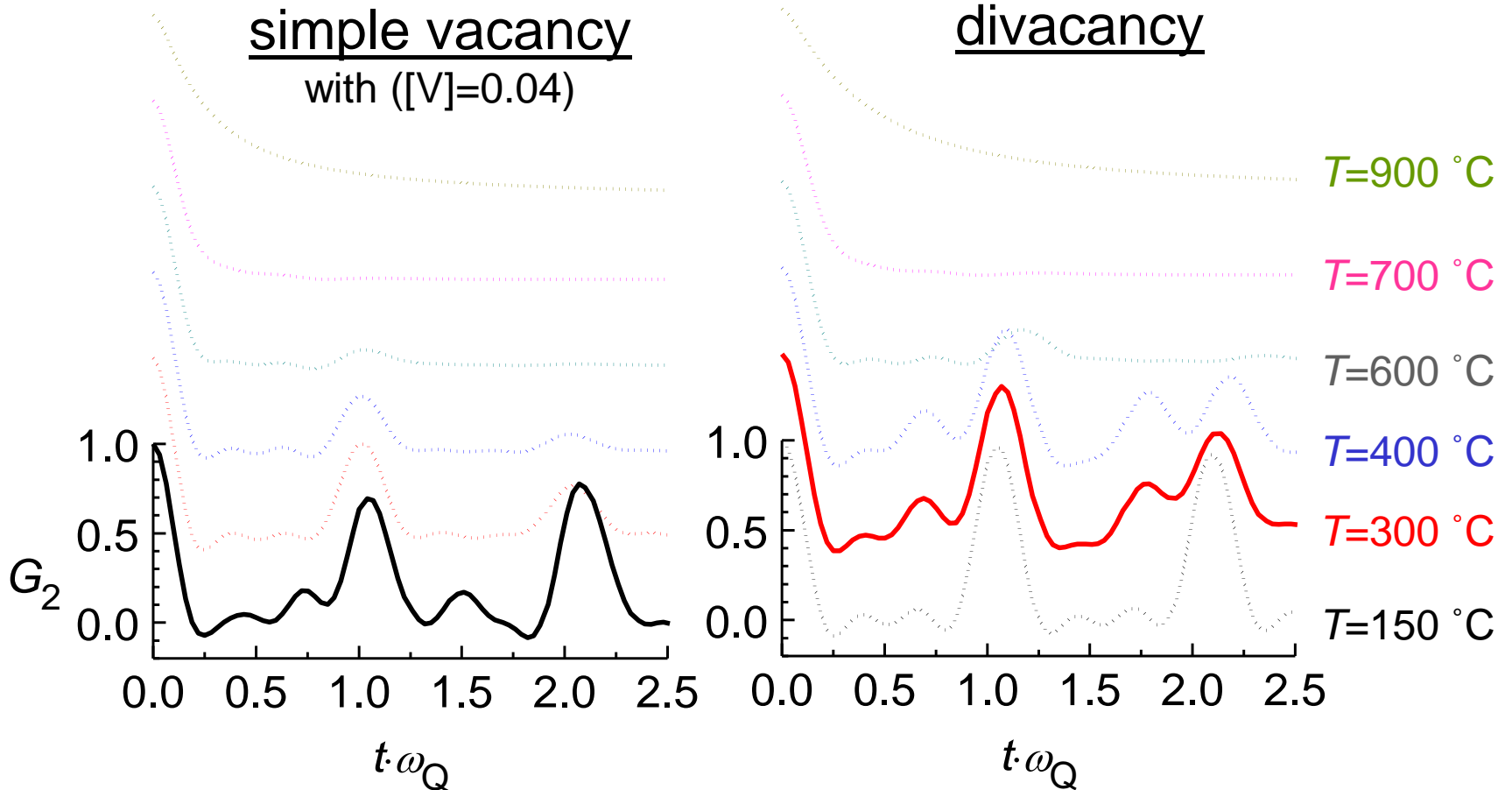
- point charge parameters for LaIn_3
- thermal vacancies with $[V] = \exp(1.4) \exp(-0.36 \text{ eV}/k_B T)$
- vacancy jump rates:
 $w_A = 1.3 \cdot 10^5 \omega_Q \exp(-0.4 \text{ eV}/k_B T)$
 $w_B = 1.3 \cdot 10^5 \omega_Q \exp(-0.5 \text{ eV}/k_B T)$



Comparison of diffusion models – part 2



Comparison of diffusion models – part 2



Summary: Modeling Complex Diffusion Mechanisms...

1. Developed a 3 step process to develop a stochastic model of fluctuating hyperfine interactions for diffusion mechanisms – applicable to any structure
2. Demonstrated that simulations of PAC spectra for simple vacancy, divacancy, and 6-jump cycle diffusion mechanisms can look different
3. Showed for a physically reasonable set of parameters that one can obtain a signal due to defects in the divacancy mechanism

Work continues to determine conditions under which one can unambiguously identify diffusion mechanisms